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### Antibacterial Corrosion Inhibitor for the Protection of Mild Steel in 1

### **M HCl Solution**

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#### Abstract

Mild steel is widely utilised as a construction material because of its mechanical qualities and low cost but its weak acid corrosion resistance limits its application. The ability of *para*-chlorobenzoylhydrazinylmethane (*P*-CBHM) as a corrosion inhibitor and antibacterial agent was investigated by gravimetric measurements, density functional theory (DFT) and antibacterial studies. The weight loss results showed that the inhibition efficiency of *P*-CBHM to prevent the corrosion of mild steel in 1.0 M HCl increased with concentration from 100 to 500 ppm, with a maximum inhibition efficiency of 96.5% at 500 ppm. The inhibitory action of the inhibitor was explained in terms of adsorption on the mild steel surface which follows a Langmuir isotherm via physisorption and chemisorption mechanisms. *P*-CBHM has significant corrosive protection efficacy and antimicrobial activity similar to the common antibacterial agent, chloramphenicol. The novelty of this work was the investigation of the protective response of *P*-CBHM which was analysed analytically on the corrosion of mild steel in 1.0 M hydrochloric acid using gravimetric and thermometric techniques and its antibacterial effects.

Keywords: Corrosion Inhibitor, DFT, gravimetric, P-CBHM, Mild Steel

#### 1. Introduction

The increasing incidence of hard-to-diagnose and multidrug-resistant diseases is a major public health concern, therefore there is an urgent need to develop new, safe and effective antimicrobial drugs. Synthesis of new chemical compounds, which may lead to broad modes of action and be less toxic to humans, is among the various approaches to identify

new targeted therapies. Hydrazine molecules are major antibacterial agents and the CO- $CH_2$ -NH-NH<sub>2</sub> part is important because of its potential biological activity (Fig. 1). Due to their high corrosion resistance, poisonous organic compounds containing N, O, and S are widely utilised as corrosion inhibitors, providing protection by preventing contact between the metallic surface and the corrosive environment through an absorption mechanism. In contrast, synthetic organic molecules are both environmentally unfriendly and expensive, limiting their utility in practical applications. Due to their costeffectiveness and environmental friendliness, green-inhibitors have received increasing attention in recent decades to overcome the constraints of standard synthetic inhibitors. To gain a better understanding of how organic compounds suppress corrosion, it is important to investigate the adsorption mechanism of these molecules on the metal surface. The efficacy of most organic corrosion inhibitors is typically related to their adsorption capabilities at the metal/solution interface, thus may be investigated using adsorption isotherms which can reveal more about the adsorption mechanism as well as the types of interactions that occur between the inhibitors and the steel surface. Indeed, corrosion inhibitors can adsorb at the interface via two types of interactions: physical adsorption (weak interactions) resulting from electrostatic interactions between charged molecules and the metal surface, and chemical adsorption resulting from electron sharing between inhibitors and d-orbitals of the iron surface.

The commonly used mild steel is particularly susceptible to corrosion in acidic solutions, therefore, using efficient strategies to impede the acidic mild steel corrosion is important [1-4]. Typically, natural inhibitors containing various functional groups like heteroatoms (nitrogen, oxygen, sulphur and phosphorous), pi-bond, in addition to polar sites are

applied to shield the mild steel surface from corrosion through developing an adsorbed layer on the metal surface as these functional groups can facilitate in the physisorption and/or chemisorption of inhibitors on the metal surface [5-9]. Unfortunately, some conventional organic inhibitors are no longer appropriate as they are harmful to the environment and living organisms [10-14], therefore scientists have been working to develop eco-friendly corrosion inhibitors [15-17] to block or retard mild steel substrate acidic corrosion [18], for example, functionalized polymer nanocomposites as anti-corrosion coatings [19]. However, due to their laborious and costly synthesis methods or short shelf life, the application of various organic inhibitors is still restricted. Consequently, the main objective of this study was to investigate the corrosion process of mild steel in 1 M hydrochloric acid solution in the absence and presence of different concentrations of a novel corrosion inhibitor, *para*-chlorobenzoylhydrazinylmethane (*P*-CBHM) (Fig. 1). The antibacterial effects of *P*-CBHM were also investigated.



Figure 1. The *P*-CBHM molecular structure.

### 2. Experimental

#### 2.1. Materials

Mild steel samples were purchased from the Company of Metal Samples and their chemical composition is shown in Table 1. The samples were cleaned according to the standard method G1-03/ASTM [20]. The corrosive solution was prepared by diluting 37% HCl (Merck-Malaysia) with distilled water.

Carbon	Manganese	Silicon	Aluminium	Sulphur	Phosphorus	Iron
0.210	0.050	0.380	0.010	0.050	0.090	balance

**Table 1.** Chemical composition of the mild steel sample (wt.%).

### 2.2. Gravimetric techniques

Hydrochloric acid (1 M) was prepared for the corrosion measurements. The mild steel stips were cut into samples measuring 4.5 mm x 2 mm x 0.5 mm, rinsed with bidistilled water and acetone, then dried in an oven. The samples were soaked in 1 M HCl media without and with the addition of *P*-CBHM (100, 200, 300, 400, and 500 ppm) at room temperature for 1, 5, 10, and 24 h. The mild steel sample mass loss was based on the total area [21-26], with the corrosion rate ( $C_R$ ), protection efficiency (IE%) and surface coverage degree ( $\theta$ ) calculated as follows [21-26]:

$$C_R = \frac{W_L}{at} \tag{1}$$

$$IE\% = \left[ \left( 1 - \frac{w_{inh}}{w_{blank}} \right) \times 100 \right]$$
(2)  
$$\theta = \left[ 1 - \frac{w_{inh}}{w_{blank}} \right]$$
(3)

### 2.3. Theoretical study

The density functional theory (DFT) with the B3LYP hybrid with the basis set 6-311<sup>++</sup>G was used to evaluate the theoretical chemical factors [30-33]. The computation characteristics of *P*-CBHM molecules cover the frontier molecular orbitals (HOMO, and LUMO), energy gap ( $\Delta E = E_{HOMO} - E_{LUMO}$ ), ionization energy (I), electronic affinity (A), electronegativity ( $\chi$ ), hardness ( $\eta$ ), softness (S), and the fraction of electron transfer ( $\Delta N$ ) which were determined according to Eqs. (4-9) [34]: I = -HOMO (4) A = -LUMO (5)  $\chi = \frac{E_{HOMO} + E_{LUMO}}{2}$ 



HOMO and LUMO are the highest occupied molecular orbital and lowest unoccupied molecular orbital, respectively.

### 2.4. Antibacterial study

The antimicrobial activity of *P*-CBHM compared to the standard medicinal medication chloramphenicol (positive control) was assessed via a disc diffusion method using the gram-negative *P. vulgaris, E. coli, P. aeruginosa* and *K. pneumonia* and gram-positive *S. aureus.* Various concentrations of *P*-CBHM were prepared using dimethylformamide

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(100, 200, 300, 400, 500 and 1000 ppm) and the pathogens were cultured in nutritional media. The *P*-CBHM was applied to 6-mm filter paper (Whatman No. 4) discs, then mounted singly and cultured in Petri dishes for 24 hours at 37°C. The antibacterial effectiveness was calculated according to [35].

#### 3. Results and discussion

#### **3.1.** Effect of concentration and immersion period

Figure 2 presents the corrosion rates ( $C_R$ ) and the inhibitory efficacies (*IE*%) of mild steel exposed to corrosive medium without and with the addition of *P*-CBHM (100, 200, 300, 400, 500 and 1000 ppm) for various immersion periods at 303 K. temperature. The  $C_R$  reduces as the *P*-CBHM concentration increases, indicating that a huge number of *P*-CBHM molecules have been adsorbed by the sample surface hence decreasing the contact between the corrosive HCl and mild steel surface. Moreover, the inhibitive efficacy increased with the increasing *P*-CBHM concentration due to the coordination bond formation between the unshared ion pairs of the oxygen and nitrogen atoms and the d-orbitals of Fe atoms on the sample surface, thereby improving the protection performance of the *P*-CBHM.

Moreover, the corrosion inhibition behaviour of *P*-CBHM (IE%) increased dramatically as the exposure period increased (Fig. 2), showing that adsorbed *P*-CBHM is continuously covering the mild steel surface. In this instance, mild steel corrosion occurs solely on the exposed sample surface or the pores of the adsorbed coating [36,37]. The corrosion rate reduced in the first 10 hours of exposure and the inhibitive efficacy increased as the immersion time increased, indicating continued *P*-CBHM adsorption on

the sample surface. C<sub>R</sub> and IE% achieved a steady state as the immersion time increased. The corrosion rate increased slightly after 24 hours of immersion, while the inhibitive efficacy reduced due to the dissociation of the protective layer on the mild steel surface. The greatest inhibition efficiency of 96.5% was achieved in the presence of 500 ppm P-CBHM, due to the P-CBHM molecules being adsorbed at the corrosion sites of the mild steel surface. The mild steel becomes inhibited and the adsorbed coating forms a protective film between the mild steel and the corrosive solution. A further increase in concentration does not improve inhibition beyond the optimum value. Indeed, the corrosion rate of mild steel decreases from 24.5 mmy-1 to 5.5 mmy-1 on the addition of 100 ppm to 500 ppm of P-CBHM, possibly due to the increased absorption and increased coverage of P-CBHM on the mild steel surface with an increasing concentration of P-CBHM. The inhibition efficiency with various immersion times from 1 h to 48 h at 500 ppm P-CBHM is shown in Figure 2. The IE% increased from 84% to 96% when the immersion time increased from 1 h to 5 h with no further change observed when the immersion time increased from 5 h to 10 h.



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Figure 2. C<sub>R</sub> and IE% plotted against immersion time.

The inhibitory efficiency of *P*-CBHM was also compared to other published inhibitors (Table 2), showing that *P*-CBHM has the highest inhibitory activity of the inhibitors listed [38-43], and performance comparable to that stated in [44, 45]. The corrosion rate decreased and the corrosion inhibition performance increased as the concentration of *P*-CBHM increased, possibly due to the increased adsorption coverage on the steel surface.

### Table 2. A comparison of the corrosive inhibitory performance of *P*-CBHM to other

Corrosion inhibitor	Metal	Acid	IE%	Ref.
para-chlorobenzoylhydrazinylmethane	Mild steel	HCl	96.5	-
N'-(2-(2-oxomethylpyrrol-1-yl)ethyl)piperidine	Mild steel	HCl	91.9	46
2-Amino-4-phenyl-N-benzylidene-5-(1,2,4-triazol-1-	Mild steel	HCl	98.1	47
yl)thiazole				
2-amino-4-phenylthiazole	Mild steel	HCl	94.7	47
1-Amino-2-mercapto-5-(4-(pyrrol-1-yl)phenyl)-1,3,4-triazole	Mild steel	HCl	96.3	48
N'-(2-hydroxybenzylidene)-2-(quinolin-8-	Mild steel	HCl	93.4	49
yloxy)acetohydrazide				
3-(4-ethyl-5-mercapto-1, 2, 4-triazol-3-yl)-1-	Mild steel	HCl	97	50
phenylpropanone				
4-ethyl-1-(4-oxo-4-phenylbutanoyl)thiosemicarbazide	Mild steel	$H_2SO_4$	88.7	51
4-benzyl-1-(4-oxo-4-phenylbutanoyl)thiosemicarbazide	Mild steel	HCl	92.5	52
4-chloro-2-((pyridin-2-ylimino)methyl)phenol	Low carbon	HCl	92.8	53
	steel			
2-N-phenylamino-5-(3-phenyl-3-oxo-1-propyl)-1,3,4-	Mild steel	HCl	95.1	54
oxadiazole				
4-ethyl-1-(4-oxo-4-phenylbutanoyl)thiosemicarbazide	Mild steel	$H_2SO_4$	88.7	55
4-ethyl-1-(4-oxo-4-phenylbutanoyl)thiosemicarbazide	Mild steel	HCl	96.1	55
4-pyrrol-1-yl-n-(2,5-dimethyl-pyrrol-1-yl)benzoylamine	Mild steel	HCl	95.8	56
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previously published organic inhibitors.

### **3.2.** Effect of temperature

The corrosion inhibition was investigated at different temperatures (303 to 333 K) as shown in Figure 3. The increase in temperature increased the  $C_R$  in the presence of *P*-CBHM, with the maximum inhibitory efficacy of 96.5% observed for 500 ppm *P*-CBHM at 303 K and 5 h immersion. The IE% reduces with increasing temperature as the generated layer (*P*-CBHM molecules) on the sample surface at 333 K has low protective ability [46, 47].



Figure 3. Inhibition efficiency plotted against temperature.

### **3.3.** Adsorption isotherms

The surface coverage ( $\theta$ ) of the *P*-CBHM molecules on the steel surface was determined based on weight loss measurements and fitted by the Langmuir isotherm (Eq. 10) as follows [48, 49]:

$$\frac{C}{\theta} = \frac{1}{K_{ads}} + C \tag{10}$$

where C represents the P-CBHM concentration and  $K_{ads}$  refers to the equilibrium constant.

Figure 4 shows the linear relationship between  $\frac{c}{\theta}$  and *C* with a regression coefficient (R2) of 0.9974, which signifies that the adsorption of *P*-CBHM on the tested surface follows the Langmuir isotherm. Moreover, the  $K_{ads}$  was determined based on the intercept to calculate the adsorption free energy ( $\Delta G_{ads}^{\circ}$ ) based on Eq. 11 [50-53]:

 $\Delta G_{ads}^{\circ} = -2.303 RT \log 55.5 K_{ads} \tag{11}$ 

Where R represents the constant of universal gas and T refers to the temperature in Kelvin.



**Figure 4.** The plot of inhibition concentration versus  $\frac{c}{a}$ .

The  $\Delta G_{ads}^{\circ}$  is  $-33.78 \text{ kJ}.mol^{-1}$ , with the negative charge referring to the spontaneous process and the  $\Delta G_{ads}^{\circ}$  from  $-40 \text{ to} - 20 \text{ kJ}.mol^{-1}$  indicates that the adsorption of *P*-CBHM on the mild steel surface occurs by physisorption and chemisorption. The physisorption mechanism suggests that the interaction accrues between the surface of mild steel and aromatic ring and/pi-bond, whereas the chemisorption mechanism occurs between unpaired elections of O and N heteroatoms in the *P*-CBHM molecule with empty iron d-orbitals on the sample surface adsorbing through interactions of the active sites and iron d-orbitals [54-56].

### **3.4.** Quantum chemical computations

The eigenvalues of the highest occupied (HOMO) and lowest unoccupied (LUMO) molecular orbitals, the HOMO–LUMO gap, electronegativity, chemical hardness, dipole moment, Fukui indices, and other parameters are the most popular molecular-electronic properties in the inhibition-efficiency correlation approach. The molecular-electronic properties of the inhibition-efficiency correlation method are based on two assumptions. The first is that these chemical characteristics are crucial reactivity markers that can be used to forecast the direction of inhibitor adsorption bonding. The higher the eigenvalue of HOMO, the greater the molecular electron donation to the metal substrate, and the lower the eigenvalue of LUMO, the greater the electron back-donation from surface states to the molecule; however, high and low imply a small HOMO-LUMO gap because ELUMO is larger than EHOMO [30-32]. The density functional theory with B3LYP at the basis set 6–31G\* was used for the quantum chemical computations (Table 3). Frontier MO energies, such as  $E_{HOMO}$  and  $E_{LUMO}$ , are crucial in predicting reactive chemical species and the ability to donate an electron is associated with E<sub>HOMO</sub> (Figure 5), thus an increase in E<sub>HOMO</sub> suggests a stronger proclivity to donate electrons to the proper acceptor with an unoccupied orbital. E<sub>HOMO</sub> facilitates the absorption of protective particles on the metal surface, with the inhibitor efficiency increased by the increased formation of an adsorbent film. The observed quantum chemistry data validates both the physisorption and chemisorption methods, as evidenced by the negative  $E_{HOMO}$  and other thermodynamic features. Since the energy gap is related to the softness and/or hardness of the inhibitor molecules, previous investigations have shown that a high  $\Delta E$  implies that

the inhibitor molecules have poor reactivity [31-33], so a soft molecule with a smaller energy gap is less reactive than a hard molecule.

The  $\Delta E$  dipole moment ( $\mu$ ) indicates a significant IE% of the inhibitor to prevent the corrosion of mild steel in 1 M HCl. Many researchers believe that heteroatoms with a strong negative charge can be adsorbed on the mild steel surface via the donors-acceptors reaction mechanism [30-34]. Moreover, a low degree of electronegativity, a large molecular weight, and a low  $\Delta E$  promote efficient adsorption of the inhibitor molecules on the MS surface, thereby reducing corrosion. The  $\Delta N$  value in Table 3 shows that the examined inhibitor molecules transport more electrons to the d-orbitals of iron atoms on the mild steel surface, thus are more efficient [30-34].



Figure 5. Optimised geometrical structure, HOMO and LUMO of P-CBHM

Computational chemic	cal P-	
parameters	CBHM	
$E_{HOMO} (eV)$	-6.569	
$E_{LUMO}(eV)$	-3.933	
$\Delta E = E_{HOMO} - E_{LUMO} \ (eV)$	2.636	
Dipole moment $(\mu)$ (D)	0.7	
Global hardness $(\eta)$	1.316	
Global softness ( $\sigma$ )	0.759	
Electronegativity $(\chi)$	5.252	
Fraction of electron transfer ( $\Delta N$ )	0.17	
hibitive mechanism		50

Table 3. Computational chemical p	parameters of the test inhibitor
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### **3.5.** Postulated inhibitive mechanism

There are various mechanisms by which organic inhibitors can prevent the corrosion of metal in an acidic environment. P-CBHM molecules can adsorb via the lone pairs in the heterocyclic rings and other functional groups to the unoccupied d-orbitals of iron atoms on the mild steel surface, with the protection efficiency depending on the chemical adsorption and physical adsorption (Figure 6). The N and O atoms in the inhibitor molecule operate as adsorption sites, with the unpaired electrons used to establish coordination bonds and chemisorb onto the mild steel surface. The protonation of nitrogen atoms is simple and can be accomplished through physisorption with chloride ions. The presence of unpaired electrons and the inductive impact of methylene groups are responsible for the prevention of corrosion. Chemisorption techniques can be utilized to characterise *P*-CBHM molecule adsorption on mild steel substrates and the activity of *P*-CBHM on the sample surface can be described based on the following equations (Eqs. 12-14):

$$Fe + P - CBHM \leftrightarrow Fe(P - CBHM)_{ads}$$
 (11)

$$Fe(P - CBHM)_{ads} \leftrightarrow Fe^{++} + P - CBHM + ne^{-}$$
(12)

$$(P - CBHM)_{aq} + H_2O_{ads} \leftrightarrow (P - CBHM)_{ads} + H_2O_{aq}$$
(13)

Donor/acceptor interactions between N and O unpaired electrons of *P*-CBHM with the empty iron d-orbitals on the sample surface allow the chemical adsorption of *P*-CBHM molecules on the sample surface. The  $\Delta G_{ads}^{\circ}$  for the tested inhibitor is  $-33.78 \ kJ \ mol^{-1}$ ,

demonstrating that the *P*-CBHM adsorption mechanism on the mild steel surface is a combination of physisorption and chemisorption [30].



**Figure 6.** The postulated mechanism of *P-CBHM* prevention of corrosion on the mild steel surface in 1 M HCl.

#### **3.6.** Antibatcterial effects

Antibacterial resistance of novel disease-causing microorganisms is increasing [57], with pathogens in the maritime environment causing infectious diseases in humans and aquaculture organisms, posing serious health risks and financial losses. Furthermore, the increased and indiscriminate use of antibiotics has led to the development of resistance to

antibiotics, as well as hypersensitivity and a loss of beneficial microorganisms in the human gut. Consequently, several bioactive and pharmacologically significant molecules have been developed as commercially available medications and there is an increased demand for new, environmentally friendly antipathogenic, antifoulant, and anticorrosion materials. The antimicrobial activity of *P*-CBHM was assessed using the disc diffusion assay (Figure 7), showing that *P*-CBHM was more effective against the gram-positive *S*. *aureus*, with the inhibitory activity increasing with concentration [57] and linked to nitrogen atoms. The *P*-CBHM demonstrated strong inhibitory activity against all tested microorganisms but was less effective than commonly used chloramphenicol.





**Figure 7.** Antimicrobial efficiency of different concentrations of *P*-CBHM compared to chloramphenicol.

#### 4. Conclusion

The new hydrazine derivative *P*-CBHM significantly inhibited the corrosion of mild steel in 1 M HCl solution due to the numerous highly efficient electronic adsorption sites including oxygen, nitrogen, chloro, and carbonyl that interacted with the iron active sites via physisorption and chemisorption mechanisms. P-CBHM also demonstrated antibacterial activity against gram-negative and gram-positive bacteria. In the conclusion, some perspective related to future research work, where the tested corrosion inhibitor will be studied using electrochemical techniques, in addition to use of other types of alloys.

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